

Table 1. Coordinates and isotropic temperature factors ( $\text{\AA}^2$ ) in  $\text{FeSnF}_6 \cdot 6\text{H}_2\text{O}$  (e.s.d.'s in parentheses)

	x	y	z	B
Fe	0	0	$\frac{1}{2}$	1.99 (2)*
Sn	0	0	0	1.67 (1)*
F	0.1460 (2)	0.1718 (2)	0.1126 (2)	4.2 (1)*
O	0.1840 (3)	0.1701 (3)	0.3813 (2)	4.8 (1)*
H(1)	0.310 (5)	0.223 (5)	0.411 (4)	6.0
H(2)	0.161 (6)	0.151 (5)	0.271 (4)	6.0

\* Equivalent isotropic temperature factor, defined as  $B_{\text{eq}} = [4/3] \sum_i \beta_i a_i^2$ , (Hamilton, 1959).

Table 2. Selected bond lengths ( $\text{\AA}$ ) and angles ( $^\circ$ ) in  $\text{FeSnF}_6 \cdot 6\text{H}_2\text{O}$  (e.s.d.'s in parentheses)

6 $\times$ Fe—O	2.116 (2)	O—H(2)	1.14 (4)
6 $\times$ Sn—F	1.945 (2)	F...H(1)	1.61 (5)
O—H(1)	1.12 (5)	F...H(2)	1.63 (4)
O—H(1)...F	176 (4)	Fe—O—H(2)	114 (2)
O—H(2)...F	165 (4)	O—Fe—O	89.96 (8)
H(1)—O—H(2)	115 (4)	F—Sn—F	89.21 (8)
Fe—O—H(1)	121 (2)		

Unlike  $\text{FeSiF}_6 \cdot 6\text{H}_2\text{O}$ , no extra reflection, signifying a lowering of symmetry from  $R\bar{3}$ , can be seen in  $\text{FeSnF}_6 \cdot 6(\text{H}_2\text{O})$ .

The structure is built up from isolated  $\text{Fe}(\text{H}_2\text{O})_6$  and  $\text{SnF}_6$  octahedra. The stability of the structure is ensured by hydrogen bonds: H atoms of water molecules point towards fluoride anions of neighbouring octahedra (Fig. 1 and Table 2). The mean Sn—F distance [1.945 (2)  $\text{\AA}$ ] is close to the sum of the ionic radii of  $\text{Sn}^{4+}$  (0.69  $\text{\AA}$ ) and  $\text{F}^-$  (1.285  $\text{\AA}$ ).

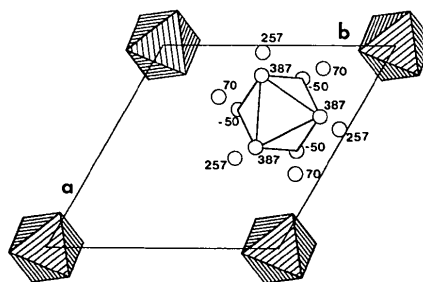


Fig. 1. [0001] projection of  $\text{FeSnF}_6 \cdot 6\text{H}_2\text{O}$ .  $\text{SnF}_6$  octahedra (at  $z = 0$ ) are shaded; heights of the H atoms of the water molecules in the  $\text{Fe}(\text{H}_2\text{O})_6$  octahedra (at  $z = 1/6$ ) are multiplied by  $10^3$ .

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## Aluminium Trimagnesium Diplatinum

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**Abstract.**  $\text{AlMg}_3\text{Pt}_2$ ,  $M_r = 490.10$ , cubic,  $Fd\bar{3}m$ ,  $a = 11.802$  (1)  $\text{\AA}$ ,  $V = 1643.9$  (4)  $\text{\AA}^3$ ,  $Z = 16$ ,  $D_x = 7.92$   $\text{Mg m}^{-3}$ ,  $\lambda(\text{Cu K}\alpha) = 1.5418$   $\text{\AA}$ ,  $\mu = 133.0$   $\text{mm}^{-1}$ ,  $F(000) = 3280$ ,  $T = 296$  K. The structure was refined to  $R = 0.017$ ,  $wR = 0.025$  for 78 absorption-corrected reflections.  $\text{AlMg}_3\text{Pt}_2$  crystallizes as an ordered ternary variant of the  $\text{NiTi}_2$  structure.

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**Experimental.** The alloy was synthesized by reaction of a stoichiometric mixture of the elements in a modified Belt-type high-pressure apparatus (Range & Leeb, 1975). This method avoids the vaporization losses of Al and Mg inevitably found in normal-pressure experiments and allows for exact control of composition. From the reaction product obtained at 40 kbar ( $\equiv 4$  GPa) and 1675 K (BN capsule, reaction time 30 min, subsequent slow cooling to 1275 K and quenching to ambient conditions) a crystal fragment

(approx. dimensions  $0.030 \times 0.040 \times 0.075$  mm) was isolated and used for data collection on an Enraf-Nonius CAD-4 diffractometer (Cu  $K\alpha$ ,  $\lambda = 1.5418$  Å, graphite monochromator in incident beam). An EDX analysis made on the same crystal after data collection showed Al, Mg, and Pt to be present in the approximate atomic ratio 1:3:2. The result of the structure analysis finally confirmed the composition  $\text{AlMg}_3\text{Pt}_2$ . Lattice parameters were refined from  $2\theta$  values of 25 reflections in the range  $6.5 \leq \theta \leq 19.8^\circ$ . Intensities were measured for  $2 \leq \theta \leq 65^\circ$ ;  $\omega$ - $2\theta$  scan technique, scan width  $(0.7 + 0.14 \tan \theta)^\circ$ . Three standard reflections showed only small random fluctuations and indicated no loss of intensity throughout data collection. Merging of the 4270 collected intensities ( $\sin \theta_{\max}/\lambda = 0.59 \text{ \AA}^{-1}$ ;  $-13 \leq h \leq 13$ ,  $-13 \leq k \leq 13$ ,  $-13 \leq l \leq 13$ ) gave 88 unique reflections ( $R_{\text{int}} = 0.088$ ), of which 78 with  $I > 0.8\sigma(I)$  were considered as observed and used for all calculations (program system *SDP3-1*; Enraf-Nonius, 1988).

The structure was solved by routine direct methods, followed by successive difference-Fourier synthesis. In the least-squares refinement  $|F|$  magnitudes were used to refine an overall scale factor, positional parameters and isotropic temperature factors. Refinement of the occupation factors for Mg, Pt, and Al confirmed the stoichiometry  $\text{AlMg}_3\text{Pt}_2$  within two standard deviations. Consequently, the s.o.f.'s were fixed at 100% before applying a numerical correction for the very serious absorption to the original 4270 data [program *DIFABS* (Walker & Stuart, 1983); correction factors minimum 0.21, maximum 1.00;  $R_{\text{int}}$  after merging of the corrected data 0.031] and performing the final anisotropic refinement (including an extinction parameter). Final  $R = 0.017$ ,  $wR = 0.025$ .  $w = 4I/[\sigma^2(I) + 0.007I^2]$ ,  $(\Delta/\sigma)_{\max} < 0.001$  in final refinement cycle, 11 variables,  $E = 1.41(4) \times 10^{-6}$ ,  $S = 0.49$ . Maximum features in the final difference-Fourier synthesis  $+0.4$ ,  $-0.3 \text{ e \AA}^{-3}$ . Atomic scattering factors and  $f'$ ,  $f''$  values as set by *SDP* program. Bond distances and angles were calculated using the program *SADIAN* (Baur & Wenninger, 1969). Final atomic coordinates and equivalent isotropic temperature factors are given in Table 1, derived bond distances in Table 2.\*

The structure of  $\text{AlMg}_3\text{Pt}_2$  is an ordered ternary variant of the  $\text{NiTi}_2$  structure (Yurko, Barton & Parr, 1959). Ni, Ti(1) and Ti(2) are replaced by Pt, Al and Mg, respectively. The Pt—Al (2.599 Å) and

\* List of structure factors, anisotropic thermal parameters and bond angles have been deposited with the British Library Document Supply Centre as Supplementary Publication No. SUP 53187 (14 pp.). Copies may be obtained through The Technical Editor, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

Table 1. Atomic coordinates and equivalent isotropic temperature factors ( $\text{Å}^2 \times 10^4$ )

$U_{\text{eq}}$  is defined as one third of the trace of the orthogonalized  $U_{ij}$  tensor.

	x	y	z	$U_{\text{eq}}$
Al	$\frac{1}{2}$	$\frac{1}{2}$	$\frac{1}{2}$	43 (8)
Mg	0.9288 (4)	$\frac{1}{8}$	$\frac{1}{8}$	42 (4)
Pt	0.71438 (3)	0.71438 (3)	0.71438 (3)	24.7 (5)

Table 2. Interatomic distances (Å)

Pt—Al	2.599 (1) ( $3 \times$ )	Al—Pt	2.599 (1) ( $6 \times$ )
Pt—Mg	2.714 (1) ( $2 \times$ )	Al—Mg	2.967 (3) ( $6 \times$ )
Pt—Mg	2.938 (3) ( $3 \times$ )	Mg—Pt	2.714 (1) ( $2 \times$ )
Pt—Pt	2.984 (1) ( $3 \times$ )	Mg—Al	2.938 (3) ( $2 \times$ )
		Mg—Pt	2.967 (3) ( $2 \times$ )

mean Pt—Mg (2.826 Å) distances agree well with those found in  $\text{Pt}_8\text{Al}_{21}$  (Range & Christl, 1988) and  $\text{Mg}_2\text{PtSi}$  (Range, Rau & Klement, 1990).

**Related literature.** No structural details for ternary Al—Mg—Pt compounds could be found in the literature. A similar structure was derived from X-ray powder data for compounds like  $\text{Nb}_3\text{Ni}_2\text{Si}$  (Gladyshevskij, Kuz'ma & Kripyakevich, 1963).  $\text{NiTi}_2$  (Duwez & Taylor, 1950) was claimed to be an oxide with composition  $\text{Ni}_2\text{Ti}_4\text{O}$  (Karlsson, 1951). Mueller & Knott (1963) concluded from powder neutron diffraction data that the structure can exist for the composition  $\text{NiTi}_2$  as well as for  $\text{Ni}_2\text{Ti}_4\text{O}$ . In the present study no evidence for the existence of additional interstitial atoms in  $\text{AlMg}_3\text{Pt}_2$  has been found.

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